

# Optical, Magnetic and Electrochemical Properties of Spinel $Mn_3O_4$ , $Co_3O_4$ Nanostructures

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## ABSTRACT

Manganese oxide ( $Mn_3O_4$ ) and Cobalt oxide ( $Co_3O_4$ ) nanocrystals were synthesized using a straightforward and cost-effective combustion method. Structural analysis through XRD, Raman Spectroscopy, and Fourier Transform Infrared Spectroscopy confirmed the tetragonal structure with  $I4_1/amd$  (141) space group for  $Mn_3O_4$  and the cubic structure with  $Fd3m$  (227) space group for  $Co_3O_4$ . The average crystallite sizes were determined using the Debye Scherer formula, resulting in approximately 14 nm for  $Mn_3O_4$  and 22 nm for  $Co_3O_4$ . SEM images displayed irregular and uniform spherical grains for  $Mn_3O_4$  and  $Co_3O_4$ , respectively, with average grain sizes of 100 nm and 120 nm. EDS and XPS spectra showed clear binding energy peaks for Manganese, Cobalt, and Oxygen, confirming the chemical purity of the samples. The optical band gap of the nanocrystals was found to be 2.93 eV for  $Mn_3O_4$  and 2.50 eV for  $Co_3O_4$ . Magnetic analysis revealed a strong ferromagnetic nature at 5 K temperature for both  $Mn_3O_4$  and  $Co_3O_4$  nanocrystals. Electrochemical investigations demonstrated that  $Mn_3O_4$  nanocrystals exhibited excellent performance as electrodes for electrochemical capacitors. At a current density of  $0.5 \text{ A g}^{-1}$ , they achieved a high specific capacitance of  $417 \text{ F g}^{-1}$ , maintaining almost 100% coulombic efficiency after 600 cycles and displaying good cyclic stability for up to 10,000 cycles. On the other hand,  $Co_3O_4$  showed inferior electrochemical behavior, indicating that  $Mn_3O_4$  nanocrystals possess superior mass transport of electrolyte ions into the electrode, making them more suitable for electrochemical capacitors.

**Keywords**—Spinel  $Mn_3O_4$  and  $Co_3O_4$  nanocrystals, band gap, specific capacitance, cyclability.

## I. INTRODUCTION

In recent times, electrochemical capacitors have emerged as promising energy storage devices due to their advantageous characteristics, such as long cycle life, high energy, and power densities [1]. Previous research indicates that the choice of electrode material and electrolyte significantly influences the performance of supercapacitors. Transition metal oxides and conducting polymers are commonly used as active electrode materials, providing high capacitance but poor cycling performance due to faradaic redox reactions [2, 3]. Alternatively, carbonaceous electrodes offer better cyclability but lower capacitance, relying on double-layer capacitance between the electrode and electrolyte.

Among transition metal oxides, some like  $Ru_2O$ ,  $MoO_3$ ,  $V_2O_5$ ,  $Fe_2O_3$ ,  $Mn_3O_4$ ,  $CuO$ ,  $ZnO$ ,  $MnO_2$ ,  $Co_3O_4$ ,  $NiO$ , and  $TiO_2$  have received extensive attention for their high theoretical specific capacitance [4, 5]. Among these,  $RuO_2$  is well-studied for its high conductivity, specific capacitance, and chemical stability.

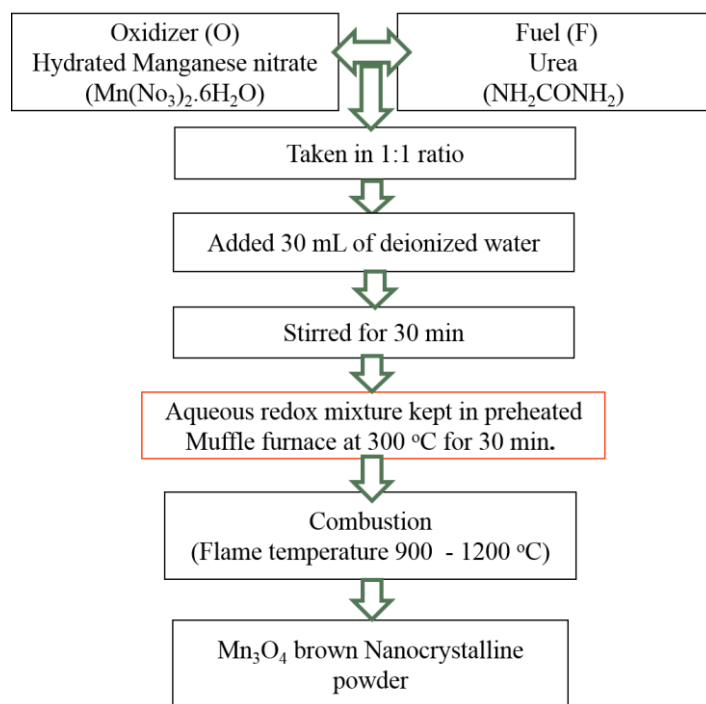
However, its limited abundance and high cost hinder practical applications. Spinel ferrites like  $\text{Mn}_3\text{O}_4$  and  $\text{Co}_3\text{O}_4$  are considered attractive electrode materials due to their cost-effectiveness, eco-friendliness, and intrinsic high capacity [6-11]. These materials find applications in various fields, including rechargeable lithium-ion batteries, molecular adsorption, electrochemistry, magnetism, and supercapacitors. Additionally, they are utilized in oxygen reduction and reaction (ORR) catalysis due to their excellent catalytic abilities in ORR. Moreover, the unique magnetic properties of these nanoparticles have garnered attention for potential applications in data storage, magnetic particle imaging, wastewater treatment, and more [13, 14].

Spinel ferrites and their composites have also found applications in diverse areas, thanks to their unusual magnetodielectric properties. For example, substituted spinel ferrites synthesized through various methods have been utilized in anti-cancer and functional radio electronic devices, catalysis, and biological applications [15-20].

However, there is limited research comparing the magnetic and electrochemical properties of spinel  $\text{Mn}_3\text{O}_4$  and  $\text{Co}_3\text{O}_4$  nanocrystals prepared via combustion synthesis. While various methods exist for synthesizing oxide nanoparticles, combustion synthesis stands out due to its simplicity, low temperature, and short processing time. This article aims to investigate and compare the optical, magnetic, ac capacitance, and electrochemical properties of spinel  $\text{Mn}_3\text{O}_4$  and  $\text{Co}_3\text{O}_4$  nanocrystals. The study reveals that  $\text{Mn}_3\text{O}_4$  nanocrystals outperform  $\text{Co}_3\text{O}_4$ , attributed to their high spin and suitable structure with a smaller crystallite size [23, 24].

## **II. EXPERIMENTAL PROCEDURE**

The reagents utilized in synthesizing spinel  $\text{Mn}_3\text{O}_4$  nanocrystals were provided by the Merck Company and possessed a purity level of over 99%. The preparation involved using manganese nitrate as the oxidant and urea as the fuel, mixed in a 1:1 molar ratio. The oxidant and fuel mixture was dissolved in 30 ml of deionized water and stirred for approximately 30 minutes using a magnetic stirrer. The resulting solution was collected in a quartz crucible and then placed inside a preheated muffle furnace set at 300 °C for a duration of 30 minutes. After the chemical reaction was complete, the final product was allowed to cool naturally to room temperature and carefully collected. The detailed step-by-step procedure for the solution combustion technique used to prepare  $\text{Mn}_3\text{O}_4$  nanocrystals is depicted in the accompanying flowchart. The same experimental procedure was replicated for the synthesis of  $\text{Co}_3\text{O}_4$  nanocrystals.



**Schematic diagram of solution combustion technique**

### A. Material Characterization

The structural characteristics of the prepared spinel  $Mn_3O_4$  and  $Co_3O_4$  nanocrystals were analyzed using X-ray power diffractometer (XRD-Model 3003 TT) and Raman spectrometer (Horiba Jobin Yvon Lab RAM HR800UV). The vibrational modes of the samples were determined using Fourier Transform Infra-red spectrometer (FTIR-ALPHA interferometer, Bruker). Morphological and elemental analysis of the samples were examined using Scanning Electron Microscopy (SEM) and Energy Dispersive Spectroscopy (EDS) (Carl ZEISS Model EVO MA15).

To perform chemical state analysis, X-ray Photo Electron Spectrometer (XPS) was utilized. The survey and core-level high-resolution spectra were measured with a monochromatic Al  $K\alpha$  X-ray (1486.6 eV) at a vacuum level of  $10^{-9}$  Torr, an applied beam current of 9 mA, and an acceleration voltage of 13 keV (117 W). The optical band gap of the nanocrystals was investigated using a UV-Vis spectrometer.

For studying the magnetic properties, Vibrational Sample Magnetometer (VSM) was employed. The electrochemical characteristics of the nanocrystals were evaluated through Cyclic Voltammetry (CV), Chronopotentiometry (CP), and Electrochemical Impedance Spectroscopy (EIS) investigations, which were performed using a 608CHI electrochemical analyzer.

### B. Electrode preparation

To fabricate the electrode, a mixture of synthesized  $Mn_3O_4$ , carbon black, and PVdF in a ratio of 80:10:10 is ground for half an hour. The resulting mixture is then combined with N-methyl-2-pyrrolidone solvent to create a slurry. Using a brush, the slurry is coated onto a chemically cleaned Nickel foam substrate. The coated electrode is subsequently dried in air at  $100\text{ }^\circ\text{C}$  for a duration of 2 hours.

For the electrochemical performance evaluation of the synthesized Mn<sub>3</sub>O<sub>4</sub> nanocrystals, a glass cell is assembled with a working electrode containing the synthesized Mn<sub>3</sub>O<sub>4</sub>, a platinum foil as the reference electrode, and Ag/AgCl as the counter electrode. The cell is filled with a 1M Na<sub>2</sub>SO<sub>4</sub> aqueous solution. The same procedure is followed for preparing the Co<sub>3</sub>O<sub>4</sub> nanocrystal electrode for further examination

### III. RESULTS AND DISCUSSION

#### A. Microstructural Properties

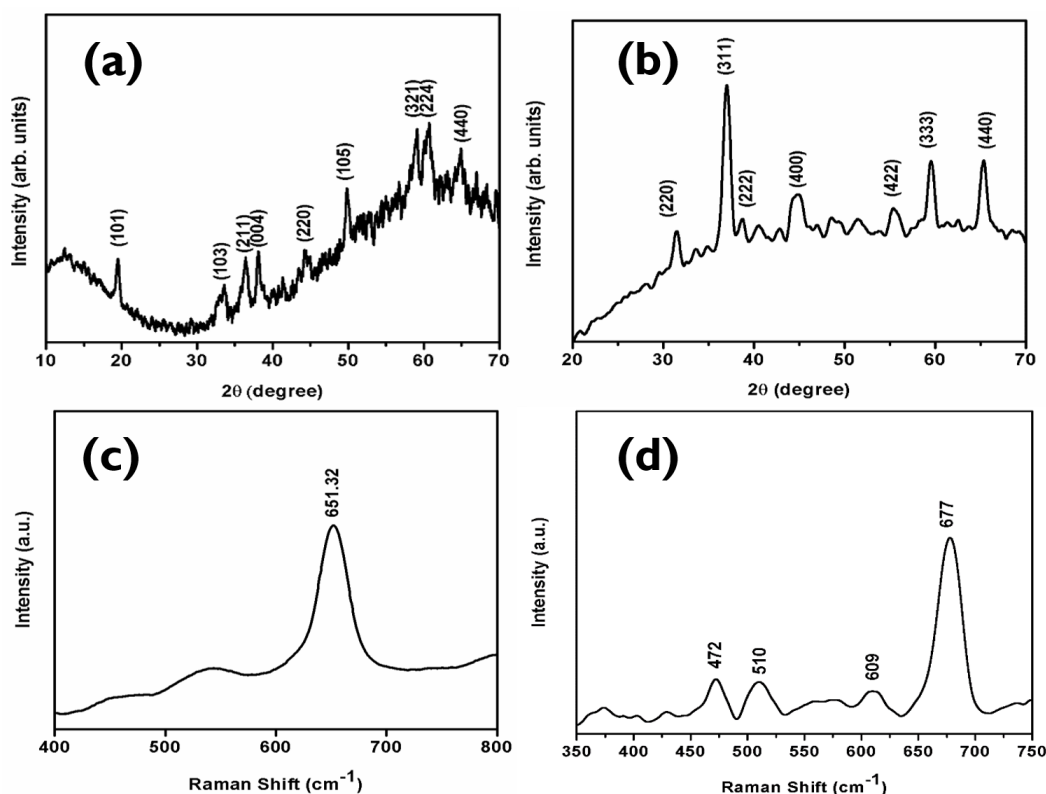
The X-ray powder diffraction (XRD) technique was employed to analyze the structure of the synthesized Mn<sub>3</sub>O<sub>4</sub> and Co<sub>3</sub>O<sub>4</sub> nanocrystals within the range of 10°-70° Bragg's angle. Figure 1a displays the XRD pattern of Mn<sub>3</sub>O<sub>4</sub>, revealing diffraction peaks at 19.43°, 33.48°, 38.18°, 44.33°, 49.83°, 59.06°, 60.68°, and 64.96°, corresponding to (101), (103), (004), (220), (105), (321), (224), and (440) reflections, respectively. Additionally, a predominant (211) orientation peak was observed at 2θ = 36.3°. These characteristic peaks confirm that Mn<sub>3</sub>O<sub>4</sub> possesses a tetragonal structure with I41/amd (141) space group, as identified by the JCPDS card No.89-4837. The lattice parameters are determined as a = b = 5.763 Å and c = 9.456 Å [15]. The high intensity and broadness of these peaks indicate a high degree of crystallinity with a relatively smaller crystallite size.

Figure 1b illustrates the XRD pattern of Co<sub>3</sub>O<sub>4</sub>, which exhibits a predominant (311) orientation peak at 2θ = 37.19°, along with other characteristic peaks at 31.54°, 38.68°, 44.59°, 55.31°, 59.55°, and 65.35° corresponding to (220), (222), (400), (422), (333), and (440) reflections, respectively. All peaks were indexed to the cubic structure with Fd3m (227) space group, indicative of a normal spinel structure, and the estimated lattice constant is a = b = c = 8.072 Å [15].

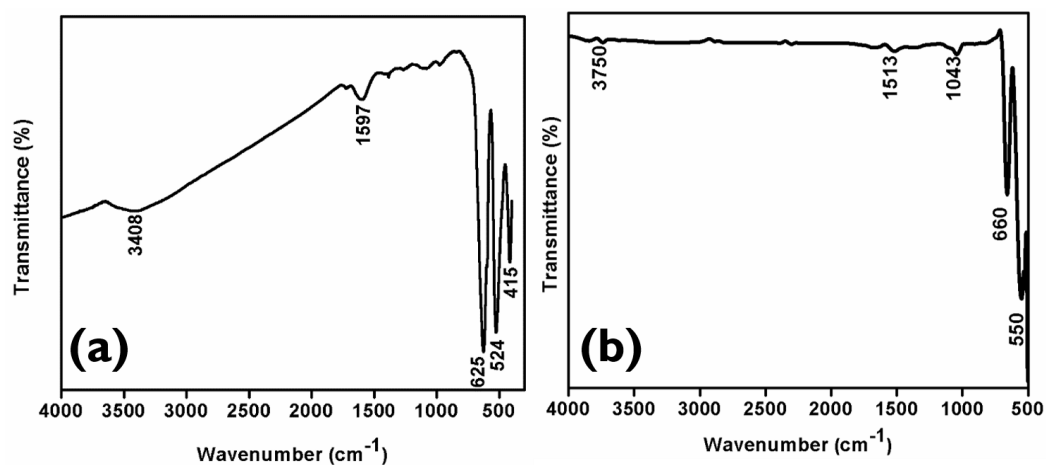
The Debye-Scherrer's formula (1) was used to determine the crystallite size (D) of the samples, resulting in sizes of 14 nm for Mn<sub>3</sub>O<sub>4</sub> and 22 nm for Co<sub>3</sub>O<sub>4</sub>, respectively

$$D = \frac{k\lambda}{\beta \cos\theta} \quad \text{--- (1)}$$

where k=0.94, λ=1.5406 Å, β=Width of the diffraction peak at half intensity in radian and θ=Bragg angle in degrees.



**Figure 1: XRD spectra and Raman spectra of spinel a, c)  $\text{Mn}_3\text{O}_4$ , b, d)  $\text{Co}_3\text{O}_4$  nanocrystals**



**Figure 2: FTIR spectrum of spinel a)  $\text{Mn}_3\text{O}_4$  b)  $\text{Co}_3\text{O}_4$  nanocrystals**

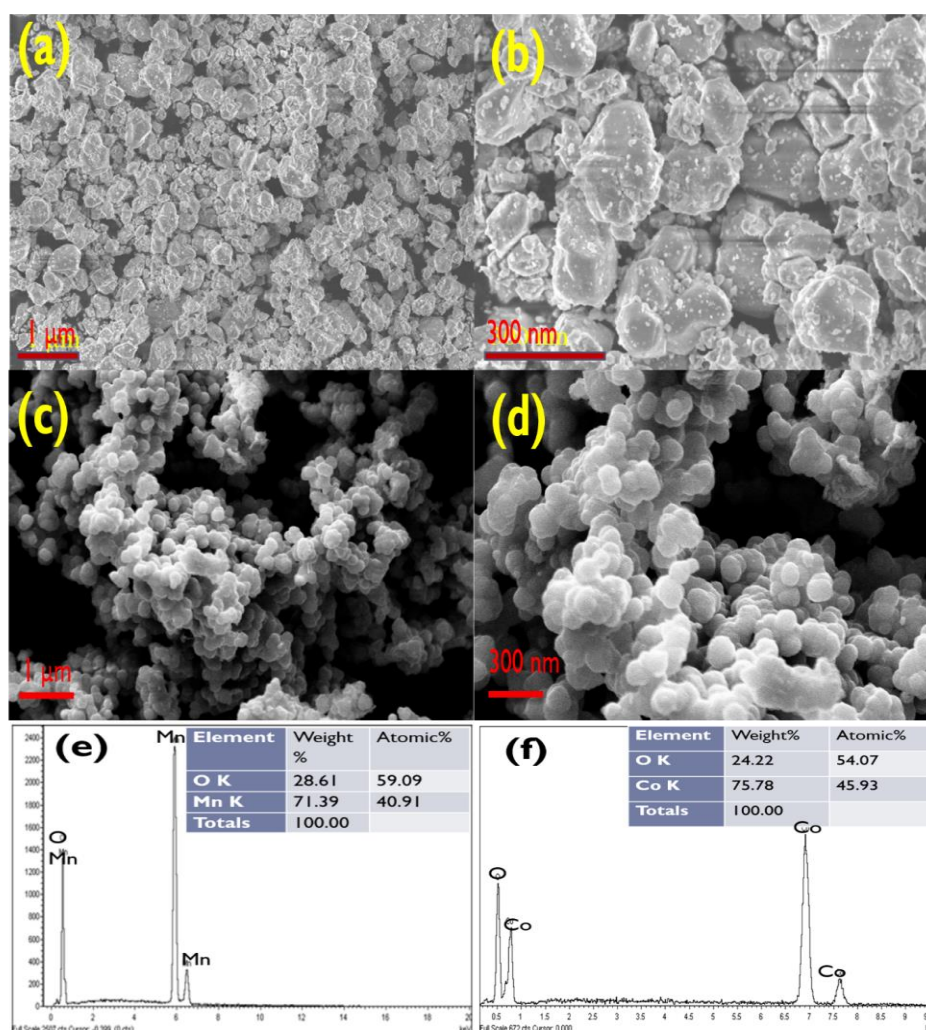
Raman spectroscopy was employed to confirm the structure of the synthesized  $\text{Mn}_3\text{O}_4$  and  $\text{Co}_3\text{O}_4$  nanocrystals. Figure 1c displays the Raman spectrum of  $\text{Mn}_3\text{O}_4$  in the wavenumber range of 400-800  $\text{cm}^{-1}$ . The peak observed at 651.32  $\text{cm}^{-1}$  corresponds to the single degenerate  $A_{1g}$  symmetry mode, which is attributed to the Mn-O stretching vibrations of  $\text{Mn}^{2+}$  in tetrahedral coordination [25]. The slight blue shift of this peak indicates a lower crystallite size, supporting the results obtained from XRD [26]. Figure 1(d) shows the Raman spectrum of  $\text{Co}_3\text{O}_4$  nanocrystals in the Raman shift range of 350-750  $\text{cm}^{-1}$ . The strong band around 670  $\text{cm}^{-1}$  and a group of weaker intensity bands around 400-600  $\text{cm}^{-1}$  are characteristic features of  $\text{Co}_3\text{O}_4$ . The strong peak at 677  $\text{cm}^{-1}$  and two weak peaks at 609 and 510  $\text{cm}^{-1}$  are attributed to  $A_{1g}$ ,  $F_{2g}$  (1), and  $F_{2g}$  (2) vibrations, respectively. These vibrations correspond to Co-O stretching vibrations, Co-O-Co asymmetric stretch, and Co-O

bending modes. Additionally, a faint band at  $472\text{ cm}^{-1}$  is identified as  $E_g$ , which is a doubly degenerate symmetry mode. These characteristic peaks confirm the cubic structure of  $\text{Co}_3\text{O}_4$  [27].

FTIR spectra of  $\text{Mn}_3\text{O}_4$  and  $\text{Co}_3\text{O}_4$  nanocrystals were recorded in the wavenumber range of  $4000\text{-}500\text{ cm}^{-1}$  to analyze the presence of vibrational bonds in the samples (Fig. 2a&b). Figure 2a exhibits absorption peaks at  $625$  and  $524\text{ cm}^{-1}$ , corresponding to the Mn-O stretching mode of tetrahedral and octahedral sites, respectively. Furthermore, a weak absorption peak at  $415\text{ cm}^{-1}$  is attributed to the vibration of manganese in an octahedral site. Vibrational bands at  $1513$  and  $3750\text{ cm}^{-1}$  are observed, corresponding to O-H bending vibrations [25]. Figure 2b shows absorption peaks at  $550$  and  $660\text{ cm}^{-1}$ , corresponding to the vibrations of  $\text{Co}^{3+}$  and  $\text{Co}^{2+}$  in octahedral and tetrahedral sites, respectively. These FTIR studies confirm the existence of Mn-O and Co-O bonds, further supporting the results obtained from Raman spectroscopy.

The morphology of  $\text{Mn}_3\text{O}_4$  and  $\text{Co}_3\text{O}_4$  nanocrystals was examined using Scanning Electron Microscopy (SEM), and the results are shown in Fig. 3. The SEM images reveal that the  $\text{Mn}_3\text{O}_4$  sample consists of irregularly shaped grains with a non-uniform distribution, having an average size of  $100\text{ nm}$  (Fig. 3a&b). Conversely, the  $\text{Co}_3\text{O}_4$  nanocrystals exhibit a uniform distribution in the form of spherical grains, with an average sphere size of  $120\text{ nm}$  (Fig. 3c&d).

To perform elemental analysis, Energy Dispersive Spectroscopy (EDS) was utilized. Figure 3e displays the binding energy peaks of Manganese and Oxygen with atomic percentages of  $40.91$  and  $59.09$ , respectively. This yields a metal-to-oxygen ratio of  $0.69$  for the  $\text{Mn}_3\text{O}_4$  sample. In Figure 3f, the binding energy peaks of Cobalt and Oxygen are shown with atomic percentages of  $45.93$  and  $54.09$ , respectively, resulting in a metal-to-oxygen ratio of  $0.85$  for the  $\text{Co}_3\text{O}_4$  sample. These EDS results confirm the chemical purity of both the  $\text{Mn}_3\text{O}_4$  and  $\text{Co}_3\text{O}_4$  samples.



**Figure 3: a & b) SEM images of spinel  $Mn_3O_4$  and c & d)  $Co_3O_4$  nanocrystals  
EDS of spinel e)  $Mn_3O_4$  f)  $Co_3O_4$  nanocrystals**

X-ray Photoelectron Spectroscopy (XPS) studies were conducted to examine the chemical state and composition of the samples. The survey spectrum of the  $Mn_3O_4$  sample contains peaks corresponding to Mn 3p, Mn 3s, O 1s, and Mn 2p, as depicted in Fig. 4a. In Fig. 4b, the Mn 2p spectrum displays binding energy peaks at 641.25 eV (Mn 2p<sub>3/2</sub>) and 653.16 eV (Mn 2p<sub>1/2</sub>), with a peak separation of 11.91 eV. Fig. 4c presents the Mn 3s spectrum, which exhibits two low binding energy peaks at 82.62 eV and 88.64 eV, with a peak difference of 5.71 eV. Furthermore, the O 1s XPS spectrum [Fig. 4d] is observed at around 532 eV, confirming the presence of oxygen in the sample. The observed binding energy differences between the Mn 3s and Mn 2p peaks are consistent with previous reports [25].

For the  $Co_3O_4$  sample, Figure 4e shows the Co 2p spectrum, which exhibits peak splitting at 780 eV (Co 2p<sub>3/2</sub>) and 795 eV (Co 2p<sub>1/2</sub>) with a peak separation of 15 eV [28]. This spectrum confirms the presence of cobalt and its chemical state in the  $Co_3O_4$  nanocrystals.

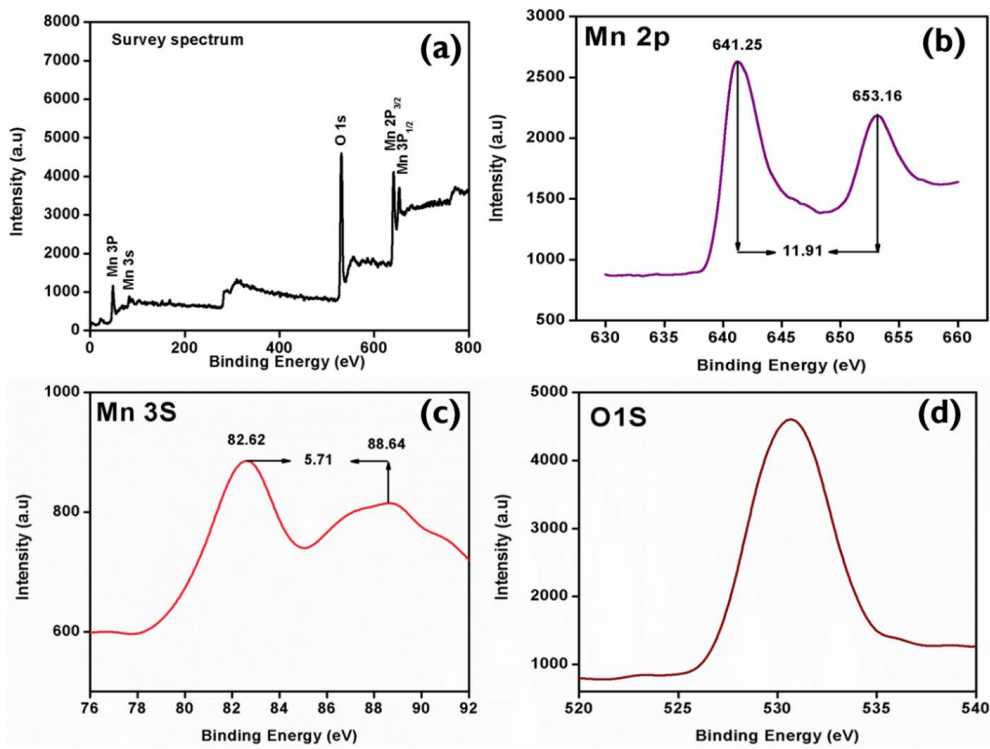


Figure 4: XPS spectra of  $\text{Mn}_3\text{O}_4$  nanocrystals: (a) Survey spectrum, (b) Mn 2p (c) Mn 3s and (d) O 1s.

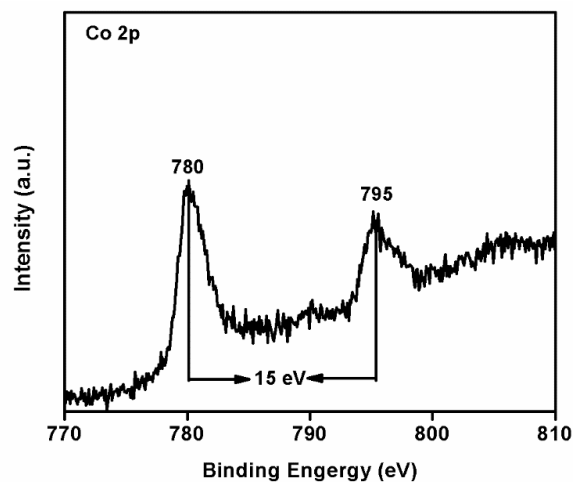


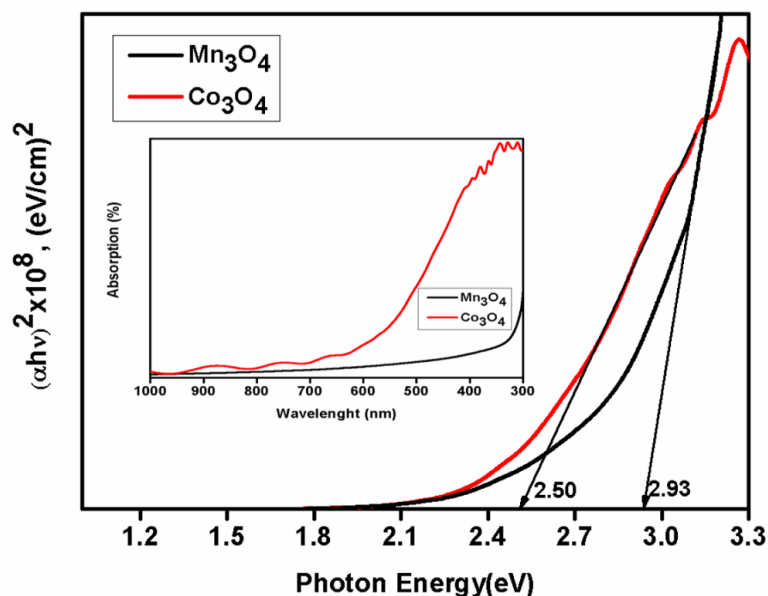
Figure 4: (e) XPS spectra of  $\text{Co}_3\text{O}_4$  nanocrystals

## B. Optical and Magnetic properties

UV-Vis spectroscopy was employed to study the optical properties of the synthesized nanocrystals in the wavelength range of 300-1000 nm, aiming to estimate their optical band gap. The resulting optical band gap spectra, along with the absorption spectra (inset of Fig. 5), were observed. The nanocrystals exhibited a p-type semiconductor nature and displayed notable absorption in the visible region. By utilizing the appropriate formula [29], the calculated optical band gap values for  $\text{Mn}_3\text{O}_4$  and  $\text{Co}_3\text{O}_4$  nanocrystals were found to be 2.93 eV and 2.5 eV, respectively.



It is noteworthy that the determined  $E_g$  values for  $Mn_3O_4$  and  $Co_3O_4$  nanocrystals in our study are higher than those reported in previous research [30]. This increase in the band gap is attributed to the quantum confinement effects and the presence of oxygen vacancies within the  $Mn_3O_4$  and  $Co_3O_4$  nanoparticles [31-34]. These factors influence the electronic structure and energy levels, leading to an elevated optical band gap in the synthesized nanocrystals



**Figure 5: Optical band gap of spinel  $Mn_3O_4$  and  $Co_3O_4$  nanocrystals**

The magnetic properties of  $Mn_3O_4$  and  $Co_3O_4$  nanocrystals were investigated using Vibrating Sample Magnetometer (VSM) over a temperature range of 5 K to 300 K, as shown in Fig. 6. The magnetic behavior of complex oxides like  $Mn_3O_4$  and  $Co_3O_4$ , with normal spinel structures, is known to strongly depend on the oxygen content. In our study, the oxygen stoichiometry of the prepared  $Mn_3O_4$  and  $Co_3O_4$  nanocrystals was determined to be  $Mn_3O_{4.06}$  and  $Co_3O_{3.9}$ , respectively, indicating oxygen excess and deficit in the respective samples. The stoichiometric  $Mn_3O_{4.06}$  sample exhibited ferromagnetic properties, as shown in Fig. 6a&b, with the magnetization curve at 5 K displaying a hysteresis loop characteristic of ferromagnetism. At 300 K, the curve resembled paramagnetic behavior.

The average oxidation level of manganese cations in the octahedral coordination of oxygen cations was found to be predominantly  $Mn^{3+}$  (70.9%) with a lesser percentage of  $Mn^{4+}$  (19.1%). The remanence magnetization ( $M_r$ ) and coercivity ( $H_c$ ) values were determined to be 0.199 emu/g and 1.06 kOe, respectively, for the  $Mn_3O_{4.06}$  sample.

On the other hand, the  $Co_3O_{3.9}$  sample displayed weak ferromagnetic properties, as shown in Fig. 6c&d, with a saturation magnetization value of 0.25 emu/g at an applied magnetic field of 0.7 kOe. The change from an antiferromagnetic state in bulk  $Co_3O_4$  to a weakly ferromagnetic state in the  $Co_3O_{3.9}$  nanocrystals can be attributed to uncompensated surface spins and/or finite size effects. The magnetic properties of nanomaterials are known to strongly depend on particle shape, size, crystallinity, and magnetization direction. The  $Co_3O_{3.9}$  sample can be regarded as a ferromagnetic matrix containing antiferromagnetic clusters due to the appearance of oxygen vacancies, resulting in antiferromagnetic interactions.

In conclusion, the magnetic behavior of  $Mn_3O_4$  and  $Co_3O_4$  nanocrystals is influenced by oxygen stoichiometry and the presence of oxygen vacancies, leading to ferromagnetic and weak ferromagnetic properties, respectively. The competition between antiferromagnetic and ferromagnetic interactions results in the formation of the spin glass state in  $Co_3O_{3.9}$  nanocrystals.

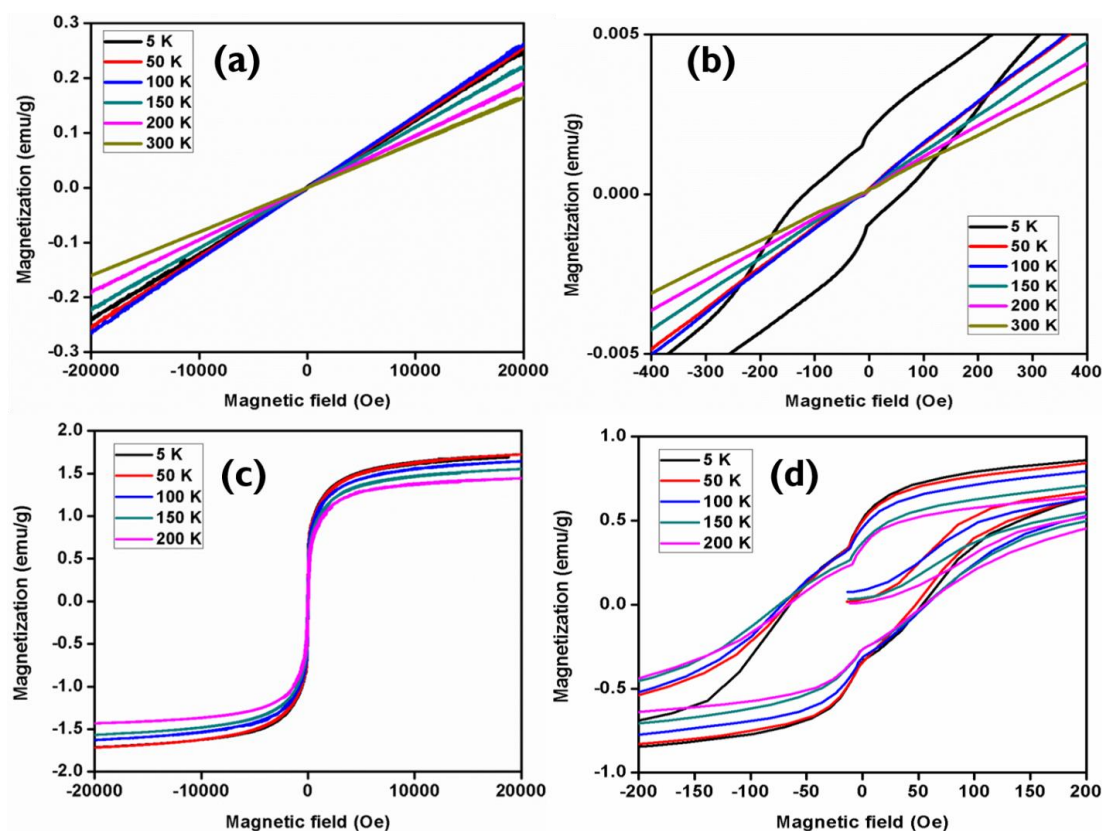
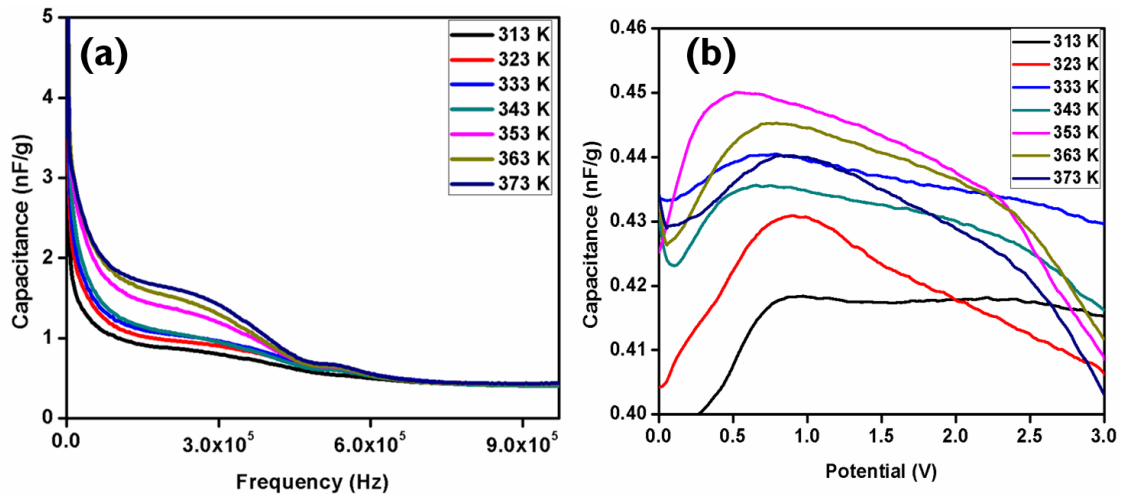


Figure 6: The M–H curves of  $Mn_3O_4$ (a & b) and  $Co_3O_4$ (c & d) nanocrystals at 5K to 300 K.

### C. Dielectric properties

The capacitive behavior of the  $Mn_3O_{4.06}$  sample is analyzed using frequency and potential dependence of capacitance curves, as depicted in Fig. 7a and 7b. Fig. 7a shows that the capacitance is higher at lower frequencies and decreases as the frequency increases, eventually reaching a constant value at higher frequencies. The high capacitance at lower frequencies can be attributed to the presence of space charge polarization, which diminishes with increasing frequency, leading to a frequency-independent behavior at higher frequencies. Moreover, the capacitance of the sample is observed to increase with temperature, as studied at different temperatures ranging from 313 K to 373 K.

Furthermore, the change in capacitance of the  $Mn_3O_{4.06}$  sample with potential at various temperatures is investigated, as shown in Fig. 7b. It is found that the decrease in oxygen vacancies leads to an increase in the conductivity of the sample, resulting in superior electrochemical properties [36].



**Figure 7: (a) Dependence of AC capacitance on frequency b) capacitance on potential of Mn<sub>3</sub>O<sub>4</sub> nanocrystals**

#### D. Electrochemical properties

The Cyclic Voltammetry (CV) studies of the nanocrystals were conducted in a 1M Na<sub>2</sub>SO<sub>4</sub> aqueous solution at different scan rates: 1-50 mVs<sup>-1</sup> for Mn<sub>3</sub>O<sub>4</sub> and 1-10 mVs<sup>-1</sup> for Co<sub>3</sub>O<sub>4</sub>, with potential window ranges of -0.1 V to +0.9 V (Mn<sub>3</sub>O<sub>4</sub>) and -0.1 V to 0.7 V (Co<sub>3</sub>O<sub>4</sub>), respectively. As shown in Fig. 8a&b, the CV curves exhibit a rectangular and semi-rectangular shape without any redox peaks, indicating the presence of interface capacitance between the electrode and the electrolyte. It was observed that the area under the curves increases with the scan rate; however, the specific capacitance decreases due to the limitation of electrolytic ions to the electrode surface at higher scan rates. Nevertheless, the shape of the CV curves remains consistent, indicating the chemical stability of the sample. The specific capacitance (C) of the samples at various scan rates was calculated using equation (2) and the values are presented in Table 1 [25, 38, 39].

Cyclic Voltammetry (CV) studies of nanocrystals are investigated in 1M Na<sub>2</sub>SO<sub>4</sub> aqueous solution at different scan rates from 1- 50 mVs<sup>-1</sup> (Mn<sub>3</sub>O<sub>4</sub>) and 1-10 mVs<sup>-1</sup> (Co<sub>3</sub>O<sub>4</sub>) in the potential window range -0.1 V to +0.9 V (Mn<sub>3</sub>O<sub>4</sub>) and -0.1 to 0.7 V (Co<sub>3</sub>O<sub>4</sub>), respectively. Fig. 8 a & b show the rectangular and semi rectangular shape of CV curves without any redox peaks indicating the interface capacitance between electrode and electrolyte. Although the area under the curves is increased with scan rate, the specific capacitance is decreased because the electrolytic ions are limited to the electrode surface at higher scan rate. At the same time, the shape of the CV curves remains same which indicates the chemical stability of the sample. The specific capacitance (C) of the samples at various scan rates measured by the following equation (2) and their values are displayed in Table 1. [25, 38, 39].

$$C = \frac{\int I(V) dV}{(2m \Delta V (V_2 - V_1))} \quad \text{---} \quad (2)$$

where,  $\int I(V) dV$  = total voltammetric charge in coulomb,  $m$  = mass of the active electrode in gram,  $\Delta V$  = scan rate in Vs<sup>-1</sup> and  $(V_2 - V_1)$  = potential window range in V. In Table 1, the specific capacitance values decrease with an increase in the scan rate. This observation is attributed to the fact that at lower scan rates, there is sufficient time for the electrolyte ions to occupy almost all the active sites of the electrode material, resulting in higher specific capacitance. Conversely, at higher scan rates, the limited time available for the electrolyte ions to reach the electrode surface leads to a lower specific capacitance.

The capacitive nature of the samples is further evident from the Chronopotentiometry studies. The charge-discharge cycles of the samples at various current densities are shown in Fig. 8c & d. The Mn<sub>3</sub>O<sub>4</sub> nanocrystals exhibit symmetric charge-discharge curves, indicating high electrochemical reversibility and excellent capacitive behavior, which corroborates the CV results. On the other hand, the charge-discharge curves of the Co<sub>3</sub>O<sub>4</sub> nanocrystals are asymmetric, with lower discharging time. The discharge capacitance (C) was calculated using the following equation (3) [25].

$$C = \frac{I \Delta t}{\Delta V m} \quad \text{---} \quad (3)$$

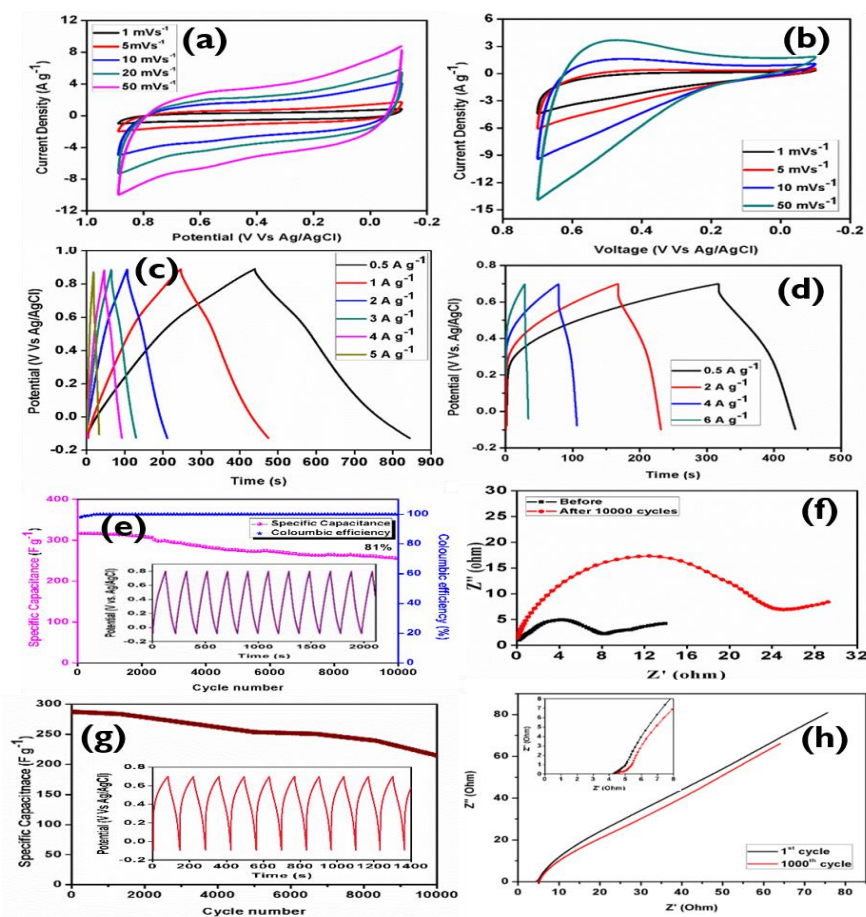
where, I= current in ampere,  $\Delta t$  = discharge time in second,  $\Delta V$ = potential window in volt and m= mass of active electrode material in gram. The Mn<sub>3</sub>O<sub>4</sub> and Co<sub>3</sub>O<sub>4</sub> nanocrystals demonstrated distinct discharge capacitance values at 0.5 Ag-1 current density, reaching 417 and 214 Fg-1, respectively. These specific capacitance values were calculated and listed in Table 2 for various current densities. The results from the constant potential (CP) studies aligned with the cyclic voltammetry (CV) findings.

A critical factor in supercapacitors is their long-term cyclic stability. To assess this, the Mn<sub>3</sub>O<sub>4</sub> charge-discharge cycles were conducted for 10000 cycles at a current density of 2 Ag-1 (Fig. 8e). Remarkably, even after 10000 cycles, 81% of capacitive retention was observed. In contrast, Co<sub>3</sub>O<sub>4</sub> exhibited 70% capacitive retention at 0.5 Ag-1 current density (Fig. 8g). The charge-discharge curves of Mn<sub>3</sub>O<sub>4</sub> displayed a symmetric pattern with cycle number, indicating a high level of electrochemical reversibility and excellent capacitive behavior [40-44]. The coulombic efficiency was calculated using formula (4).

$$\eta = \frac{t_d}{t_c} \times 100 \quad \text{---} \quad (4)$$

where,  $\eta$ ,  $t_d$  and  $t_c$  refers to coulombic efficiency, discharge time and charging time respectively. The Mn<sub>3</sub>O<sub>4</sub> electrode exhibited 100% coulombic efficiency after 600 cycles as shown in Figure 8e.

Electrochemical Impedance Spectroscopy (EIS) is another characterization technique employed to investigate the capacitive behavior of the electrode materials. EIS analysis of the sample is conducted in the frequency range from 1 Hz to 1 MHz to corroborate the CV results. The Nyquist plots of the Mn<sub>3</sub>O<sub>4</sub> nanoparticles are shown in Fig. 8f. The intersection of the Nyquist plots on the real axis at higher frequencies represents the solution resistance (RS). It is observed that both the solution and charge transfer resistances of the materials are smaller before cycling and become higher after 10,000 cycles, which leads to a decrease in specific capacitance with the number of cycles, as depicted in Fig. 8f&h.



**Figure 8:** a) CV curves of spinel  $\text{Mn}_3\text{O}_4$  and b)  $\text{Co}_3\text{O}_4$  nanocrystals at different scan rates. c) Charge-discharge curves of spinel  $\text{Mn}_3\text{O}_4$  and d)  $\text{Co}_3\text{O}_4$  nanocrystals at different scan rates e) Specific capacitance and coulombic efficiency with cycle number f) Nyquist plots of  $\text{Mn}_3\text{O}_4$  nanocrystals g) Specific capacitance with cycle number h) Nyquist plots of  $\text{Co}_3\text{O}_4$  nanocrystals.

#### IV. CONCLUSION

The spinel nanostructures of  $\text{Mn}_3\text{O}_4$  and  $\text{Co}_3\text{O}_4$  were successfully synthesized using the solution combustion technique. XRD and Raman studies confirmed the establishment of tetragonal  $\text{Mn}_3\text{O}_4$  and cubic  $\text{Co}_3\text{O}_4$  structures, with estimated crystallite sizes of 14 nm and 22 nm, respectively. FTIR measurements further supported the formation of manganese oxide and cobalt oxide, evident from the presence of Mn-O and Co-O bonding. The SEM micrographs showed dispersed irregular grains of approximately 100 nm size for  $\text{Mn}_3\text{O}_4$  and uniformly distributed spherical grains with an average size of 120 nm for  $\text{Co}_3\text{O}_4$ . The optical band gaps of the samples were measured to be 2.93 eV for  $\text{Mn}_3\text{O}_4$  and 2.50 eV for  $\text{Co}_3\text{O}_4$  nanocrystals. Additionally, the magnetic properties of both samples exhibited strong ferromagnetic behavior at around 5 K temperature. The electrochemical characterization studies, including CV, Chronopotentiometry, and EIS, were performed in 1M  $\text{Na}_2\text{SO}_4$  aqueous electrolyte. The as-prepared  $\text{Mn}_3\text{O}_4$  nanocrystals demonstrated the highest specific capacitance of  $417 \text{ F g}^{-1}$  at a current density of  $0.5 \text{ A g}^{-1}$ , along with good cyclic stability and almost 100% coulombic efficiency. Based on these results, it can be concluded that nanocrystalline  $\text{Mn}_3\text{O}_4$  holds promise as an excellent electrode material for supercapacitor applications.

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